

Constitutive equations

- Scalar microstress ϖ conjugate to $\dot{\mathbf{d}}$

$$\varpi = \underbrace{\frac{\partial \hat{\psi}_{\mathbf{R}}(\mathbf{\Lambda})}{\partial \mathbf{d}}}_{\varpi_{\text{en}}} + \underbrace{\alpha + \zeta \dot{\mathbf{d}}}_{\varpi_{\text{diss}}},$$

with $\alpha = \hat{\alpha}(\mathbf{\Lambda})$ and $\zeta = \hat{\zeta}(\mathbf{\Lambda})$ positive-valued scalar functions.

- Vector microstress $\boldsymbol{\xi}$ conjugate to $\nabla \dot{\mathbf{d}}$

$$\boldsymbol{\xi} = \frac{\partial \hat{\psi}_{\mathbf{R}}(\mathbf{\Lambda})}{\partial \nabla \mathbf{d}}.$$

This is taken to be energetic, with no dissipative contribution.

Governing pdes

1. Equilibrium equation:

$$\text{Div } \mathbf{T}_R + \mathbf{b}_{0R} = \mathbf{0},$$

where \mathbf{b}_{0R} is a non-inertial body force.

2. Microforce balance:

The microforces ϖ and ξ obey the balance,

$$\text{Div } \xi - \varpi = 0.$$

This microforce balance, together with the thermodynamically consistent constitutive equations for ϖ and ξ gives the evolution equation for the damage variable d ,

$$\hat{\zeta}(\mathbf{\Lambda}) \dot{d} = \left\langle -\frac{\partial \hat{\psi}_R(\mathbf{\Lambda})}{\partial d} + \text{Div} \left(\frac{\partial \hat{\psi}_R(\mathbf{\Lambda})}{\partial \nabla d} \right) - \hat{\alpha}(\mathbf{\Lambda}) \right\rangle.$$

Since ζ is positive-valued, the right hand side of the equation above must be positive for \dot{d} to be positive and the damage to increase monotonically.

Boundary and initial conditions

1. Boundary conditions for the pde governing the evolution of χ :

$$\left. \begin{aligned} \chi &= \check{\chi} && \text{on } \mathcal{S}_\chi \times [0, T], \\ \mathbf{T}_R \mathbf{n}_R &= \check{\mathbf{t}}_R && \text{on } \mathcal{S}_{\mathbf{t}_R} \times [0, T]. \end{aligned} \right\}$$

2. Boundary conditions for the pde governing the evolution of d :

$$\begin{aligned} \dot{d} &= 0 && \text{on } \mathcal{S}_d \times [0, T], \\ \nabla d \cdot \mathbf{n}_R &= 0 && \text{on } \partial B \setminus \mathcal{S}_d \times [0, T]. \end{aligned}$$

The initial data is taken as

$$\chi(\mathbf{X}, 0) = \mathbf{X}, \quad \text{and} \quad d(\mathbf{X}, 0) = 0. \quad \text{in } B.$$

Specialization of the constitutive equations

- Replace chain stretch with distortional effective stretch and allow for slight compressibility:

$$\bar{\lambda} = \sqrt{\frac{\text{tr } \bar{\mathbf{C}}}{3}} \quad \text{distortional effective stretch} \quad J = \det \mathbf{F} \quad \text{volumetric Jacobian}$$

- Entropy density: $\eta_R = \hat{\eta}_R(\bar{\lambda}, \lambda_b) = -Nk_B n \left[\left(\frac{\bar{\lambda} \lambda_b^{-1}}{\sqrt{n}} \right) \beta + \ln \left(\frac{\beta}{\sinh \beta} \right) \right] \quad \beta = \mathcal{L}^{-1} \left(\frac{\bar{\lambda} \lambda_b^{-1}}{\sqrt{n}} \right)$

- N number of chains per unit reference volume
- n number of links in each chain

- Internal energy density: $\hat{\varepsilon}_R(\lambda_b, J, d, \nabla d) = (1 - d)^2 \hat{\varepsilon}_R^0(\lambda_b, J) + \hat{\varepsilon}_{R, \text{nonloc}}(\nabla d).$

$$\hat{\varepsilon}_R^0(\lambda_b, J) = \frac{1}{2} \bar{E}_b (\lambda_b - 1)^2 + \frac{1}{2} K (J - 1)^2,$$

$$\bar{E}_b = N n E_b \quad \text{— net bond stiffness}$$

$$K \quad \text{— bulk modulus}$$

$$\hat{\varepsilon}_{R, \text{nonloc}}(\nabla \mathbf{d}) = \frac{1}{2} \varepsilon_R^f \ell^2 |\nabla \mathbf{d}|^2,$$

$$\varepsilon_R^f = N n \varepsilon_b^f \quad \text{— net bond dissoc. energy}$$

$$\ell \quad \text{— intrinsic length scale}$$

- Bond deformation stretch still given by local free energy minimization

$$\frac{\partial \psi_R}{\partial \lambda_b} = 0.$$

Evolution equation for damage variable

- Scalar microforce:

$$\varpi = 2(1 - d)\varepsilon^0(\lambda_b, J) + \varepsilon_R^f + \zeta \dot{d}$$

$$\varepsilon_R^f \stackrel{\text{def}}{=} N n \varepsilon_b^f \quad \text{net dissociation energy} \quad \zeta > 0 \quad \text{viscous regularization parameter}$$

- Vector microforce: $\xi = \varepsilon_R^f \ell^2 \nabla d$

- Microforce balance, $\text{Div } \xi - \varpi = 0$: $\zeta \dot{d} = 2(1 - d)\varepsilon^0(\lambda_b, J) - \varepsilon_R^f + \varepsilon_R^f \ell^2 \nabla d$

- To enforce $d \in [0, 1]$: $\zeta \dot{d} = 2(1 - d) \left\langle \hat{\varepsilon}_R^0(\lambda_b, J) - \varepsilon_R^f / 2 \right\rangle - \varepsilon_R^f [d - \ell^2 \Delta d]$,
and to account for the irreversible nature of chain scission $\dot{d} \geq 0$, we introduce

$$\mathcal{H}(t) \stackrel{\text{def}}{=} \max_{s \in [0, t]} \left\langle \hat{\varepsilon}_R^0(\lambda_b(s), J(s)) - \varepsilon_R^f / 2 \right\rangle .$$

- Then the evolution equation for d may be written as

$$\zeta \dot{d} = \left\langle 2(1 - d)\mathcal{H} - \varepsilon_R^f [d - \ell^2 \Delta d] \right\rangle .$$

Governing pdes

Equilibrium equation:

$$\text{Div } \mathbf{T}_R + \mathbf{b}_R = \mathbf{0}$$

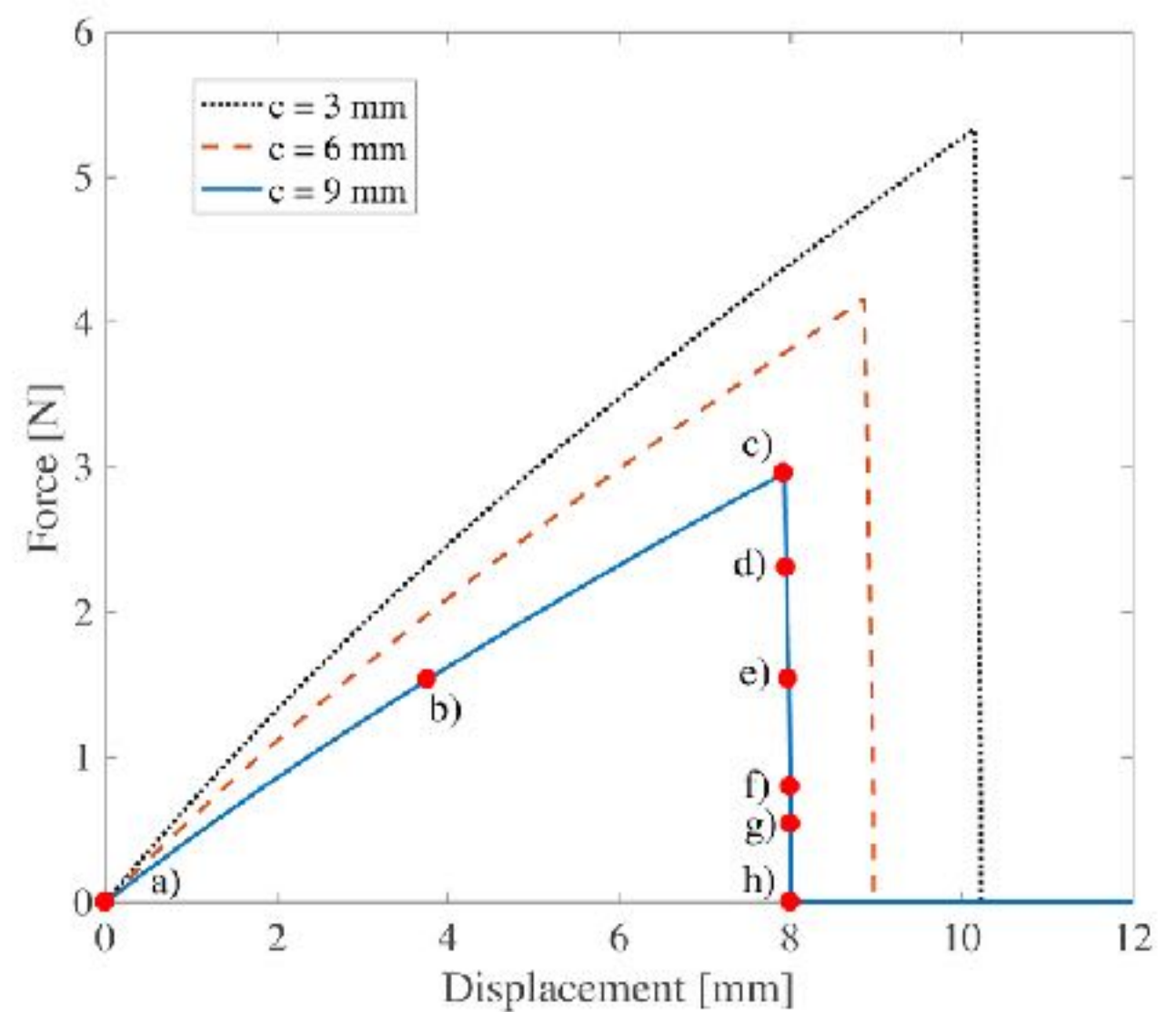
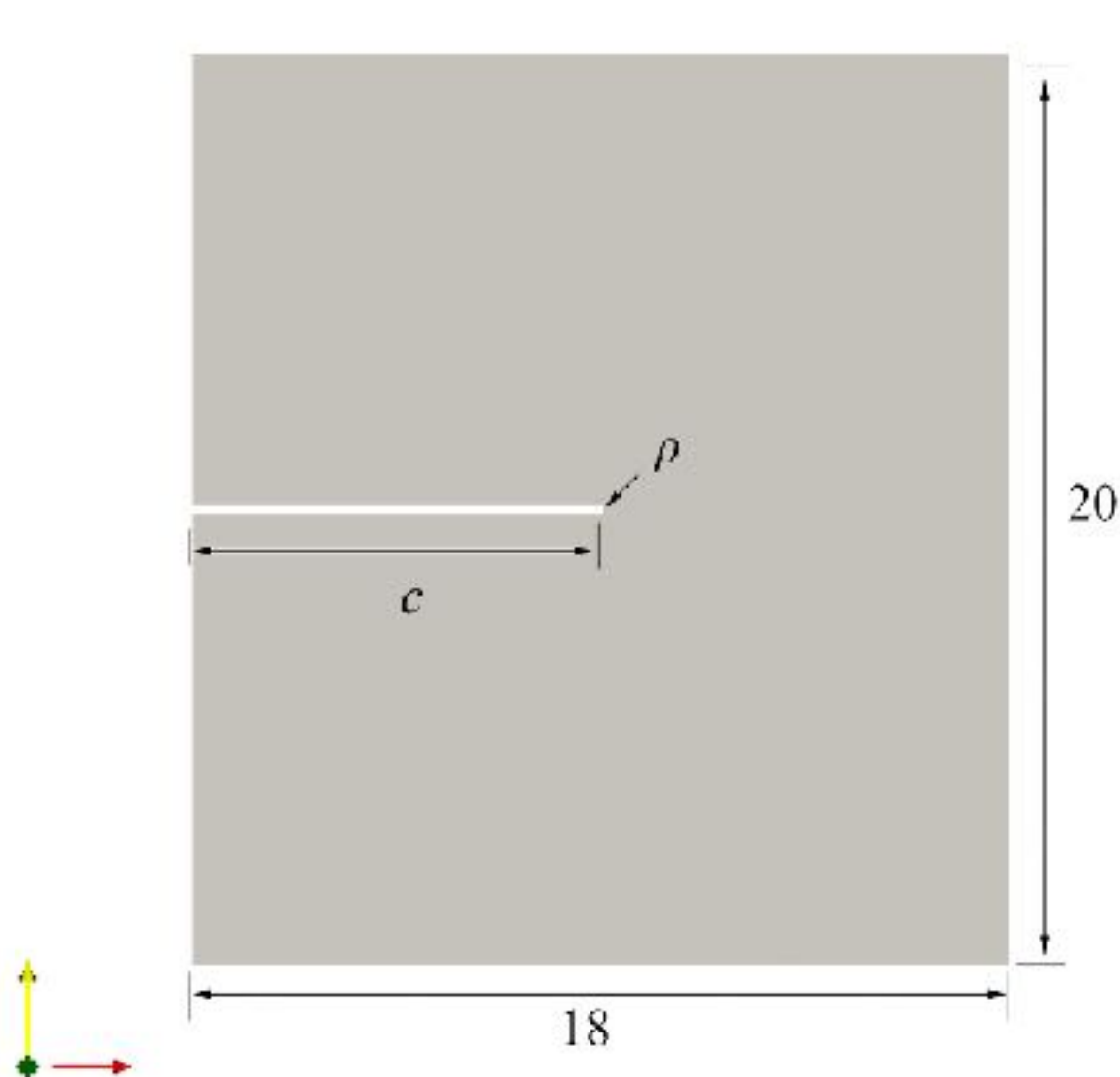
Evolution equation for d :

$$\zeta \dot{d} = \langle 2(1 - d)\mathcal{H} - \varepsilon_R^f [d - \ell^2 \Delta d] \rangle$$

$$\mathcal{H}(t) \stackrel{\text{def}}{=} \max_{s \in [0, t]} \langle \hat{\varepsilon}_R^0(\lambda_b(s), J(s)) - \varepsilon_R^f / 2 \rangle$$

+ BCs and ICs

Plane-stress simulation of single-edge-notch fracture of an elastomeric sheet



Dimensions in mm

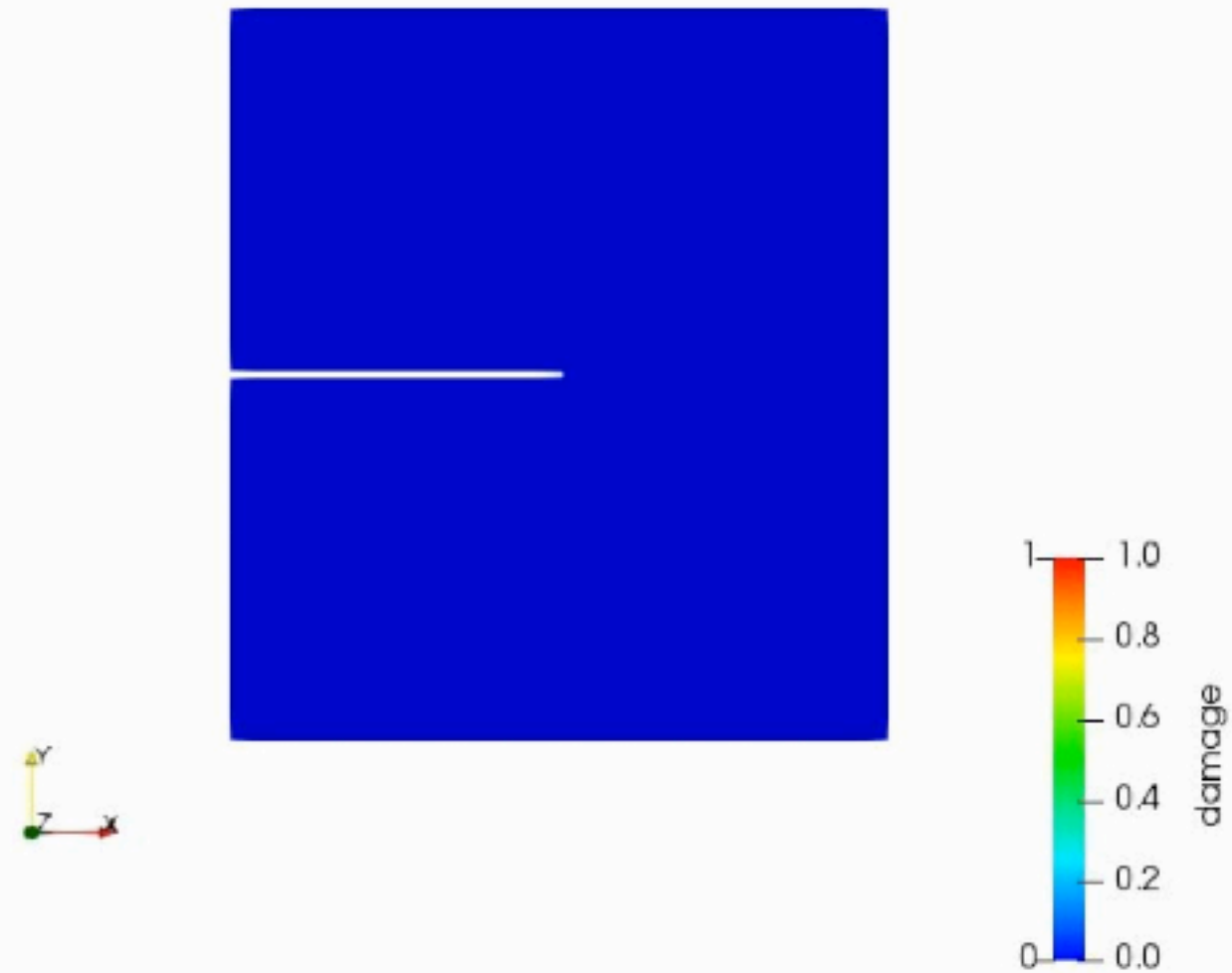
- Thickness: 1mm; $\rho = 0.1$ mm
- Stretch rate $1 \times 10^{-3}/s$

Material parameters

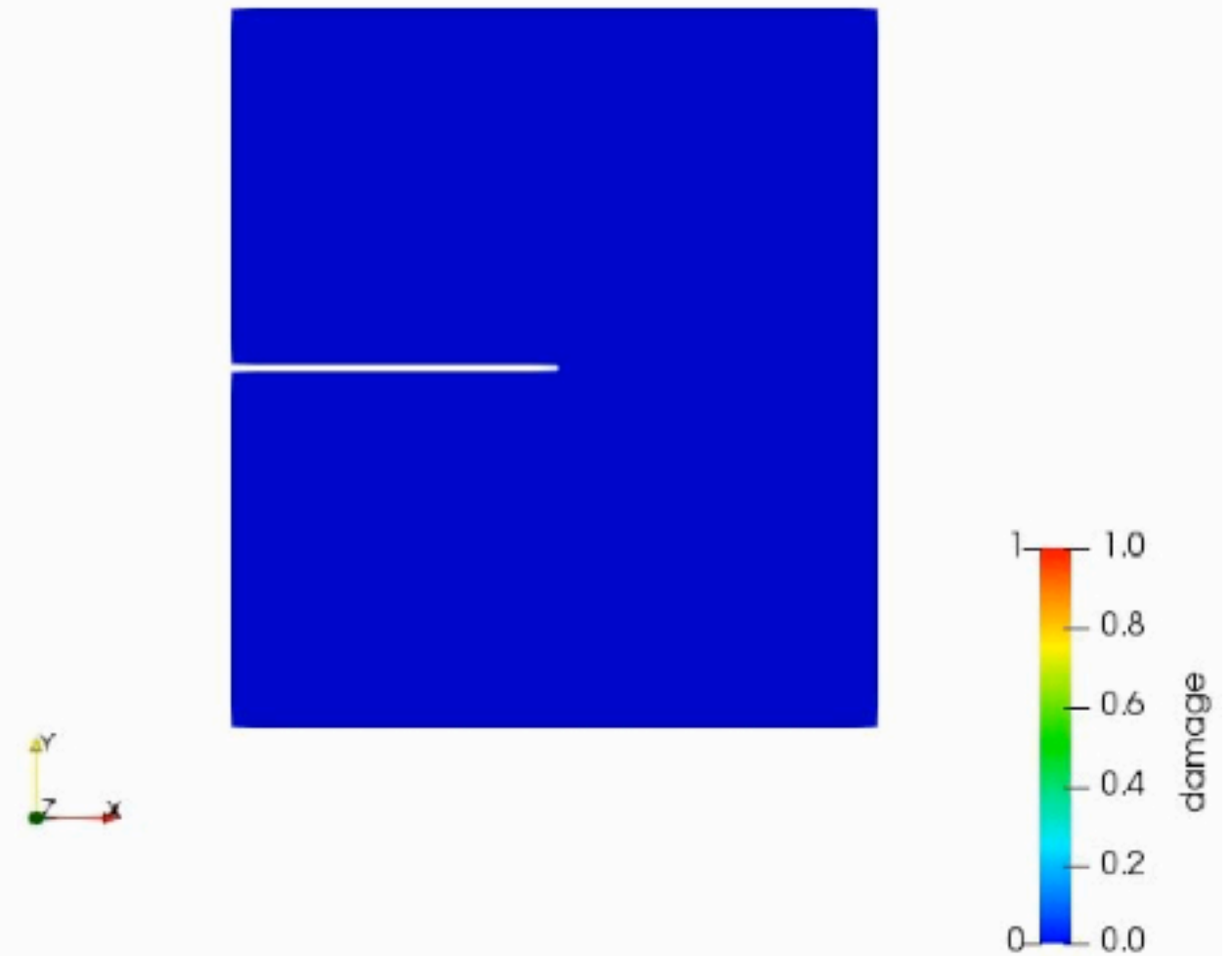
$G_0 = Nk_B\vartheta$	n	$\bar{E}_b = NnE_b$	K	$\varepsilon_R^f = Nn\varepsilon_b^f$	ℓ	ζ
0.25 MPa	4	5 MPa	5 MPa	2.5 MJ/m ³	100 μ m	20 kPa·s

Plane stress simulation of single-edge-notch fracture of an elastomeric sheet

Contours of damage
with highly damaged elements
shown

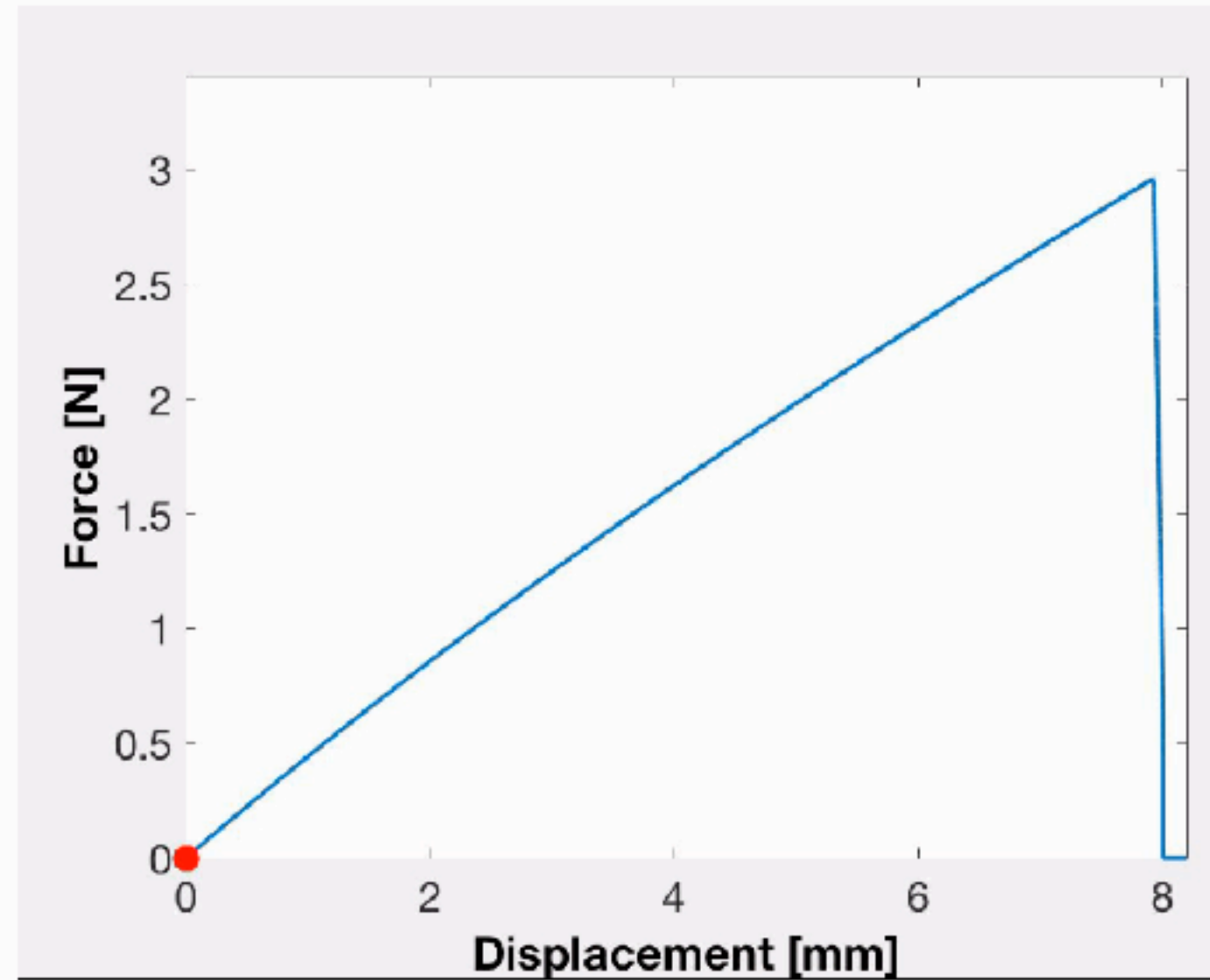
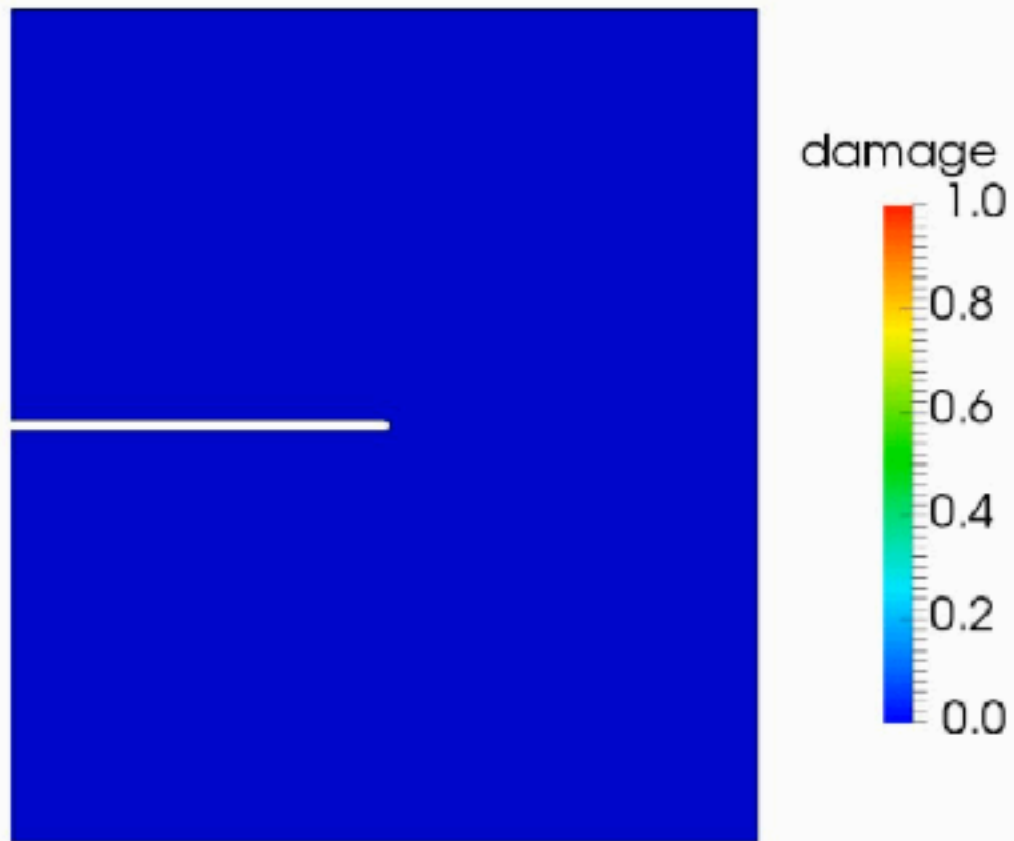


Contours of damage
without highly damaged elements

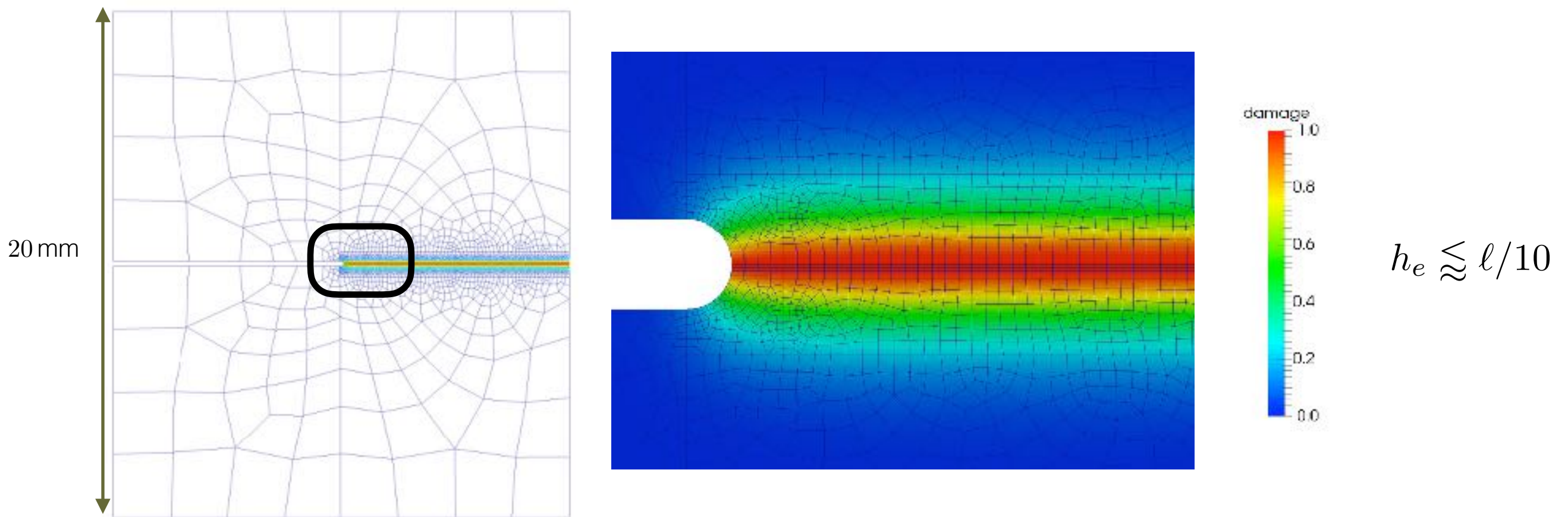


Single-edge-notch fracture of an elastomeric sheet

Contours of damage



The length scale ℓ and mesh for the single-edge-notched specimen



- Actual values of ℓ in elastomeric materials are expected to be $\ell \approx 1\mu\text{m}$. For $h_e \approx \ell/10$ — so that $h_e \approx 100\text{nm}$, which is *exceedingly small*.
- For modeling macroscopic-dimensioned specimens, several mm in length, for pragmatic reasons we consider ℓ to be regularization parameter for the gradient-damage theory.
- Corresponding to a small but computationally-tractable mesh size h_e selected for macroscopic-dimensioned specimens, a suitably large value of $\ell = 100\mu\text{m}$ has been chosen, and the value of ε_R^f suitably reduced so that

$$\varepsilon_R^f \times \ell \approx G_c,$$

where G_c is the value of experimentally-measured macroscopic critical energy release rate for a given material.